

Multifunctional biophotonic nanostructures inspired by the longtail glasswing butterfly for medical devices

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Numerous living organisms possess biophotonic nanostructures that provide colouration and other diverse functions for survival. While such structures have been actively studied and replicated in the laboratory, it remains unclear whether they can be used for biomedical applications. Here, we show a transparent photonic nanostructure inspired by the longtail glasswing butterfly (*Chorinea faunus*) and demonstrate its use in intraocular pressure (IOP) sensors in vivo. We exploit the phase separation between two immiscible polymers (poly(methyl methacrylate) and polystyrene) to form nanostructured features on top of a Si_3N_4 substrate. The membrane thus formed shows good angle-independent white-light transmission, strong hydrophilicity and anti-biofouling properties, which prevent adhesion of proteins, bacteria and eukaryotic cells. We then developed a microscale implantable IOP sensor using our photonic membrane as an optomechanical sensing element. Finally, we performed in vivo testing on New Zealand white rabbits, which showed that our device reduces the mean IOP measurement variation compared with conventional rebound tonometry without signs of inflammation.

During their lifetime an estimated 8–10% of Americans (5–6% in other developed nations) depend on implantable medical devices to support bodily functions^{1,2}. Consequently, efforts to develop medical implant technologies are increasing. A major deterrent to these efforts, however, has been the requirement to incorporate multiple functionalities within a tightly constrained footprint while ensuring acceptable in vivo performance and reliability^{3–6}. Inspiration for engineering multifunctional surfaces is often drawn from nature, which boasts a plethora of nanostructures with a wide array of desirable properties^{4–8}. For example, vertically tapered needle-like nanostructures found on the wings of insects exhibit multifunctionality including omnidirectional antireflection, self-cleaning, antifouling and bactericidal properties^{9–13}. Such properties may prove to be advantageous for biomedical applications such as in vivo sensing, imaging and stimulation.

Herein, we seek inspiration from the multifunctional biophotonic nanostructures found on the transparent wings of the longtail glasswing butterfly (*Chorinea faunus*) to advance the versatility of micro-optical implants whose practical use is often limited by the angle dependency of sensing and readout processes^{14,15} as well as short- and long-term biofouling^{15–17}. We characterized in detail the surface and optical properties of the short-range-ordered nanostructures found on *C. faunus* wings, which could overcome the shortcomings of micro-optical implants. We reveal that *C. faunus* relies on relatively moderate-aspect-ratio (aspect ratio ≈ 1) chitin nanostructures to produce (1) transparency that is a unique combination of wavelength-selective anti-reflection and angle-independent transmission resulting from isotropic Mie scattering, and (2) antifouling properties through disruption of cellular growth similar to that observed on high-aspect-ratio (aspect ratio > 1) structures found in nature^{12,13}. Drawing our inspiration from the *C. faunus*

nanostructures, we created low-aspect-ratio (aspect ratio < 1) bio-inspired nanostructures on freestanding Si_3N_4 -membranes using a highly scalable phase-separation-based polymer-assembly process. Unlike previous high-aspect-ratio bio-inspired nanostructures replicating antireflection^{9,12,13}, we engineered the pseudo-periodic arrangement and dimensions of nanostructures to control isotropic scattering and enhance omnidirectional optical transmission, which could benefit sensing and readout processes of micro-optical implants. In addition, improving on the anti-biofouling properties of high- and moderate-aspect-ratio nanostructures that typically rely on physical cell lysis^{12,13,18}, we engineered low-aspect-ratio nanostructures to generate strong nanostructure-mediated hydrophilicity and an anti-adhesion barrier for proteins and cellular fouling without inducing cell lysis and inflammation.

To demonstrate the medical multifunctionality of bio-inspired nanostructures, we combined bottom-up nanofabrication with top-down microfabrication processes to yield a nanostructured micro-optical implant that senses intraocular pressure (IOP) for diagnosis and management of glaucoma, a leading cause of irreversible blindness^{19–21}. We confirmed significant improvement in the bio-inspired nanostructured sensor's optical readout angle, pressure-sensing performance and biocompatibility during a one-month in vivo study conducted in rabbits.

Multifunctional nanostructures of *C. faunus*

C. faunus (Fig. 1a) belongs to the Riodinidae family found in South America. *C. faunus* wings are distinct from most other transparent wings in nature^{9,11,22}. They have a rare combination of two transparent regions that transmit light differently: basal transparent areas close to the thorax (indicated by a blue arrow in Fig. 1a); and post-discal transparent areas further away from the thorax (red arrow

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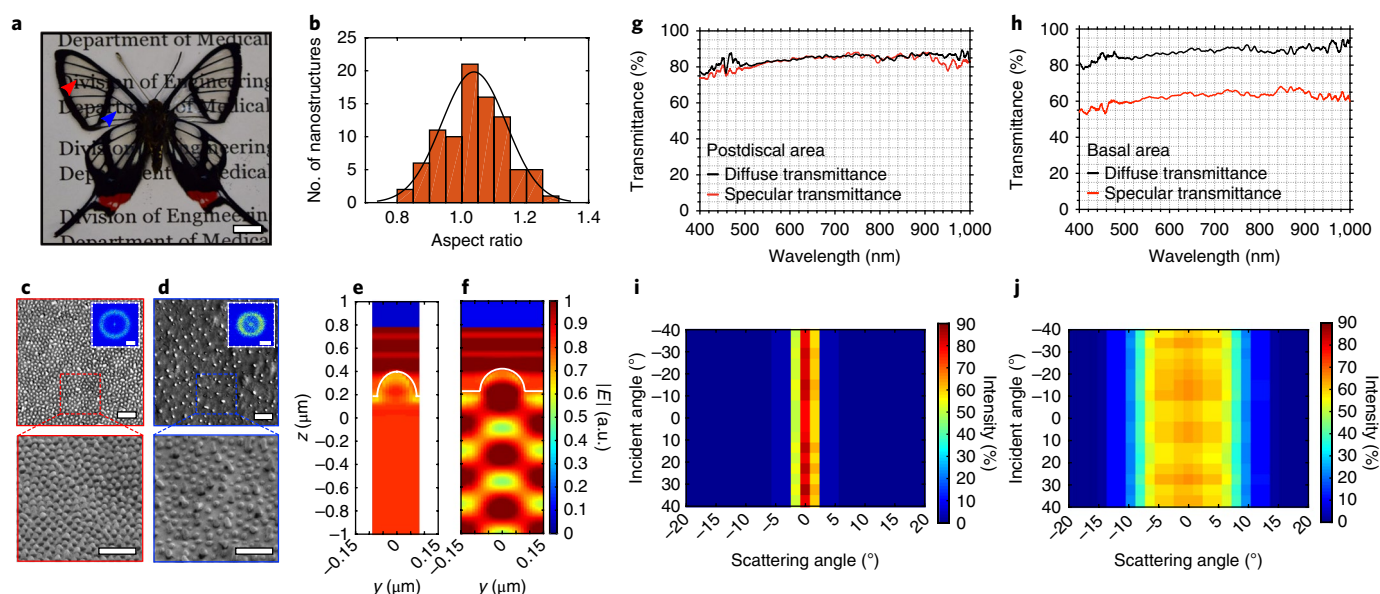


Fig. 1 | Characterization of *C. faunus* wings. **a**, Photo of a *C. faunus* butterfly under visible light. The red and blue arrowheads indicate the postdiscal and basal areas, respectively. Scale bar, 1 cm. **b**, Statistical analysis of the aspect ratio (height over base diameter) of nanostructures on the *C. faunus* wing. An average aspect ratio of 1.090 ± 0.041 is estimated by fitting a Gaussian profile. **c,d**, SEM images of the densely packed postdiscal area (**c**) and sparsely located basal area (**d**) dome-shaped nanostructures. Insets: 2D Fourier transform of the corresponding nanostructures. Scale bars, 1 μm (inset: 2 μm^{-1}). **e,f**, Finite-difference time-domain simulations of the near-field scattering profile for the postdiscal area (**e**, cell periodicity: 150 nm) and basal area (**f**, cell periodicity: 300 nm) at a wavelength of 420 nm. **g**, Measured diffuse and specular transmittance of the postdiscal area; difference in spectra within the experimental uncertainty. **h**, Measured diffuse and specular transmittance of the basal area showing a 20% difference in transmittance and a noticeable scattering property. **i,j**, The forward scattering of postdiscal (**i**) and basal (**j**) areas were recorded for a range of incident angles varying from -20 to 20° at a wavelength of 420 nm. The measurements depict the different degree of the haze effect for the transparent (low in haze) and translucent (high in haze) areas. The postdiscal area exhibits specular transmittance with a low scattering angle of $\pm 3^\circ$. Conversely, the basal area scatters light in a forward direction with a much wider scattering angle of $\pm 12^\circ$.

in Fig. 1a) of both the forewing and hindwing. High-resolution scanning electron microscopy (SEM) of the postdiscal transparent area reveals dome-shaped nanopillars with moderate aspect ratios (1.090 ± 0.041 ; Fig. 1b,c and Supplementary Fig. 1) compared with other natural transparent wings with higher aspect ratios (>1)^{10–12}. Interestingly, the basal transparent area is composed of similarly shaped nanostructures at a lower density (Fig. 1d). Two-dimensional fast Fourier transforms of the SEM images showed ring-shaped distributions (insets of Fig. 1c,d), which confirmed their short-range-ordered arrangements^{11,23}. The finite diameters of the rings in *k*-space quantified the average periods of the nanostructures as 140–180 nm and 200–300 nm for the postdiscal and basal areas, respectively.

The zone-dependent variation in average interstructural periods on the *C. faunus* wing plays an important role in the extent of light scattering on the wing. The postdiscal area with an average period below 200 nm remains scattering-free and anti-reflective in the visible–near-infrared (VIS–NIR) regime, and this is well-explained by the effective medium theory and the transfer-matrix model as in other anti-reflective subwavelength nanostructures found in nature^{11,24} (Supplementary Fig. 2). The basal area with an average period exceeding 200 nm comparable to light wavelengths shows forward narrow-angle scattering due to more sparsely located, moderate-aspect-ratio, low-index nanostructures^{25,26}. Our finite-difference time-domain simulations performed on both groups of nanostructure at a wavelength of 420 nm produced matching results that confirmed the scattering mechanism of the basal area (Fig. 1e,f). Though we used the same structural height and diameter for both groups in the simulations, the nanostructures with a 150 nm period (similar to the postdiscal area) did not alter the

transmitted field (Fig. 1e), whereas nanostructures with a 300 nm period (similar to the basal area) showed forward scattering of the transmitted light (Fig. 1f). The scattering phenomenon of the basal region is moreover confirmed by the difference observed between the specular transmittance and total transmittance in the VIS–NIR range shown in Fig. 1g,h.

To further analyse the transmissive scattering properties of both areas, we performed angle-resolved scattering spectroscopy in the VIS range and varied both the incident and detection angles (Fig. 1i,j). Having almost identical specular and diffuse transmissions in the VIS–NIR range (Fig. 1g), the postdiscal area exhibited specular transmittance with a low scattering angle of $\pm 3^\circ$ (Fig. 1i). Contrastingly, the basal area scattered light in the forward direction with scattering angles up to $\pm 12^\circ$ and showed negligible changes with incident angle variation, demonstrating its potentially very useful angle-independent scattering property (Fig. 1j). This scattering property could ameliorate the difficulty of detecting optical signals at wide angles, a commonly observed challenge among many light-based devices¹⁴ such as implantable IOP sensors¹⁵. (See Supplementary Section 1 and Supplementary Figs. 3 and 4 for more details on the biological significance of the multifunctional transparency and its dual nano-structural basis present on *C. faunus* wings.)

Additionally, the periods of the nanostructures on *C. faunus* wings also influence the wetting properties, and the static contact angles in the postdiscal and basal areas measured 105° and 85° , respectively. The contact angle in the postdiscal area is larger due to the higher surface roughness²⁷. In our experiments, these nanostructures with moderate aspect-ratios, similar to nanostructures with high aspect-ratios, resist microbial and cellular growth (Supplementary Figs. 5 and 6).

Development of bio-inspired nanostructured membranes

Inspired by the nanostructures on the basal area of *C. faunus* wings, we implemented short-range-ordered nanostructures on Si_3N_4 -membranes (Fig. 2a) by utilizing a highly scalable bottom-up fabrication process based on polymer phase separation^{28,29}. Si_3N_4 was chosen for its ease of fabrication on Si and proven performance as an optically transparent and mechanically robust freestanding membrane in microdevices^{30,31} as well as for its intrinsic hydrophilicity, which is crucial to the antifouling property of the nanostructures as discussed in more detail later in this work.

We created disc-shaped nanostructures with aspect ratios ranging from 0.15 to 0.90 and performed parametric studies to determine optical and anti-biofouling properties (Supplementary Fig. 7). Nanostructures with an aspect ratio of 0.45 (Fig. 2b), smaller than the aspect ratio observed in the nanostructures found on *C. faunus* wings (≈ 1), were found to provide an optimal balance between the anti-biofouling and angle-independent optical properties best-suited for optical implants (Supplementary Section 2). Henceforth, the aspect ratio of nanostructures integrated on the membrane is 0.45 unless stated otherwise.

The SEM image of the nanostructured Si_3N_4 -membrane is shown in Fig. 2c. The 2D fast Fourier transform of the SEM image shown in the inset indicates a short-range order with a mean period of 445 ± 60 nm, similar to the periodicity of the basal area. The contact

angle on the surface of the nanostructured Si_3N_4 -membrane was 17° , suggesting an increase in hydrophilicity compared to 38° measured on flat Si_3N_4 without nanostructures (Supplementary Fig. 8).

We characterized the optical properties of the nanostructured Si_3N_4 -membrane using angle-resolved transmission spectroscopy in the VIS–NIR range and compared the results to a flat Si_3N_4 -membrane without nanostructures (Fig. 2d,e). Using the nanostructures, the angle independence of the Si_3N_4 -membrane transmission was improved by 50%. 3D simulation of the fabricated structures (Supplementary Fig. 9b) further confirms the improved angle-independent transmittance. This angle-independent transmission results from the isotropic nature of the forward scattering caused by the short-range-ordered nanostructures, which is irrespective of the incident angle (Fig. 1j). As the total transmission is a combination of ballistic (specular) transmission through the thin membrane and scattered transmission caused by the nanostructures (Supplementary Fig. 10)³², the angle-independent property of the scattered component decreases the overall angle dependence of the total transmission.

Biophysical properties of the nanostructured surface

In vitro testing compared the adhesion of representative proteins, prokaryotes and eukaryotes on nanostructured and flat Si_3N_4 surfaces with lysine-coated glass slides as positive controls. Flat Si_3N_4 is

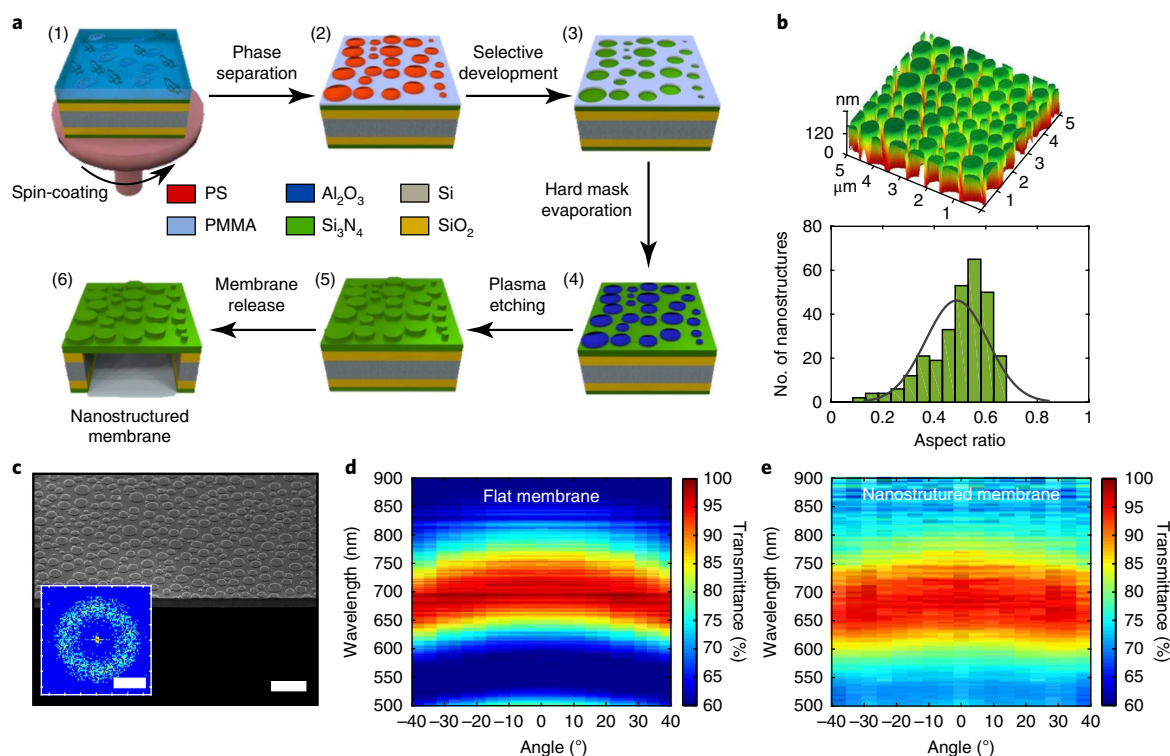


Fig. 2 | Nanostructured Si_3N_4 -membrane fabrication and optical properties. **a**, Fabrication flow of the bio-inspired nanostructured Si_3N_4 -membrane: (1) spin-coating of a blend solution of PMMA and PS in MEK on $\text{Si}_3\text{N}_4/\text{SiO}_2/\text{Si}/\text{SiO}_2/\text{Si}_3\text{N}_4$ wafer; (2) phase separation of the polymers; (3) selective development of the PS; (4) Al_2O_3 hard mask formation using evaporation and lift-off processes; (5) nanopillar-pattern transfer onto Si_3N_4 using plasma etch; and (6) release of the nanostructured Si_3N_4 -membrane using backside optical lithography and reactive ion etching. **b**, 3D atomic force microscopy image of the nanostructured Si_3N_4 -membrane and nanostructure aspect ratio of 0.450 ± 0.065 approximated with a Gaussian fit. **c**, SEM image of the nanostructures on the Si_3N_4 -membrane and corresponding ring-shaped 2D Fourier power spectrum shown inset. Scale bars, $0.5 \mu\text{m}$; inset $1.25 \mu\text{m}^{-1}$. **d**, Experimentally obtained angle-resolved total transmittance of flat Si_3N_4 -membrane showing a transmission peak around 705 nm due to the light interference introduced by the thin membrane with its peak blue-shifted 30 nm at 40° incident angle due to the angle-dependent nature of the coherent interference process, which agrees with analytical thin-film modelling (Supplementary Fig. 9a). **e**, Experimentally obtained angle-resolved total transmittance of nanostructured Si_3N_4 -membrane, showing significant reduction in angle-dependence. The integration of nanostructures on the Si_3N_4 -membrane broadens the total transmission-peak profile, moves its centre from 705 to 685 nm, and limits the magnitude of the peak shift to 15 nm at 40° , compared with 30 nm for a flat Si_3N_4 -membrane.

moderately hydrophilic (contact angle: 35–40°) and known to vigorously promote cell adhesion and proliferation due to increased adsorption of proteins when compared to more hydrophilic surfaces (contact angle: <20°)^{33,34}. Hence, we further increased the hydrophilicity of the Si₃N₄ surface by varying the aspect ratios of the nanostructures from 0.15 to 0.90 and systematically controlled surface hydrophilicity (Supplementary Fig. 8). Once strong hydrophilicity is achieved (contact angle: <20°), a nanostructure-mediated aqueous barrier forms on the surface and limits protein adsorption and cell adhesion to provide an anti-adhesion property (Supplementary Section 3, Supplementary Figs. 11 and 16)^{6,35,36}.

We initially investigated the surface adhesion of two representative proteins: fluorescent-labelled bovine serum albumin for its cardinal role in blood–material interactions⁴ and high non-specific binding affinity to the surfaces of biomaterials³⁷; and streptavidin for its specific binding affinity to Si₃N₄ surfaces³⁸. Fluorescence-intensity-based quantification of the adhesion force (Figs. 3a and b and Supplementary Fig. 11) demonstrated adhesion on flat Si₃N₄ surfaces was three and two times greater than nanostructured Si₃N₄ surfaces for albumin and streptavidin, respectively.

We then quantified bacterial adhesion using *Escherichia coli* transformed with the green fluorescent protein (Supplementary Fig. 12). In addition to being a popular prokaryotic model, *E. coli* was chosen for its pathogenic potential to cause Gram-negative and often antibiotic-resistant infections on and around implants^{39,40}.

Bacteria on each surface were quantified through a measure of colony-forming units (CFU) (Fig. 3c) and fluorescence-intensity measurements (Supplementary Fig. 13). Both results indicated significantly lower bacterial adhesion on the nanostructured surface compared to flat Si₃N₄. Additionally, the SEM image of individual bacterial cells on the nanostructured surface shows no disruption to their shape, indicating no physical lysis (Supplementary Fig. 14).

The HeLa cell line was chosen as a representative eukaryote for its proven robustness, aggressive growth rate and adherent nature, which prompts its frequent usage in adhesion and cytotoxicity assays^{41,42}. After 72 h, the adherent cell density on the flat Si₃N₄ was eight times greater than that on the nanostructured Si₃N₄ surface (Figs. 3d and e). Next, a mortality ratio, the number of dead cells to the number of living cells, was computed for each surface every 24 h over a 72-h period. The difference in the mortality ratios of the two surfaces after 72 h was not statistically significant (Supplementary Fig. 15), which suggested the nanostructured surfaces inhibited eukaryote adhesion and proliferation without inducing cell death.

These results highlight the advantage of the anti-biofouling approach based on strong hydrophilicity and anti-adhesion properties (Supplementary Figs. 11–16). High- or moderate-aspect-ratio nanostructures either with tapered sharp tips or dome-shaped tips as in *C. faustus* display potent geometry-dependent bactericidal properties that induce large stresses and deformation on cell walls regardless of their surface chemical composition⁴³ and actively

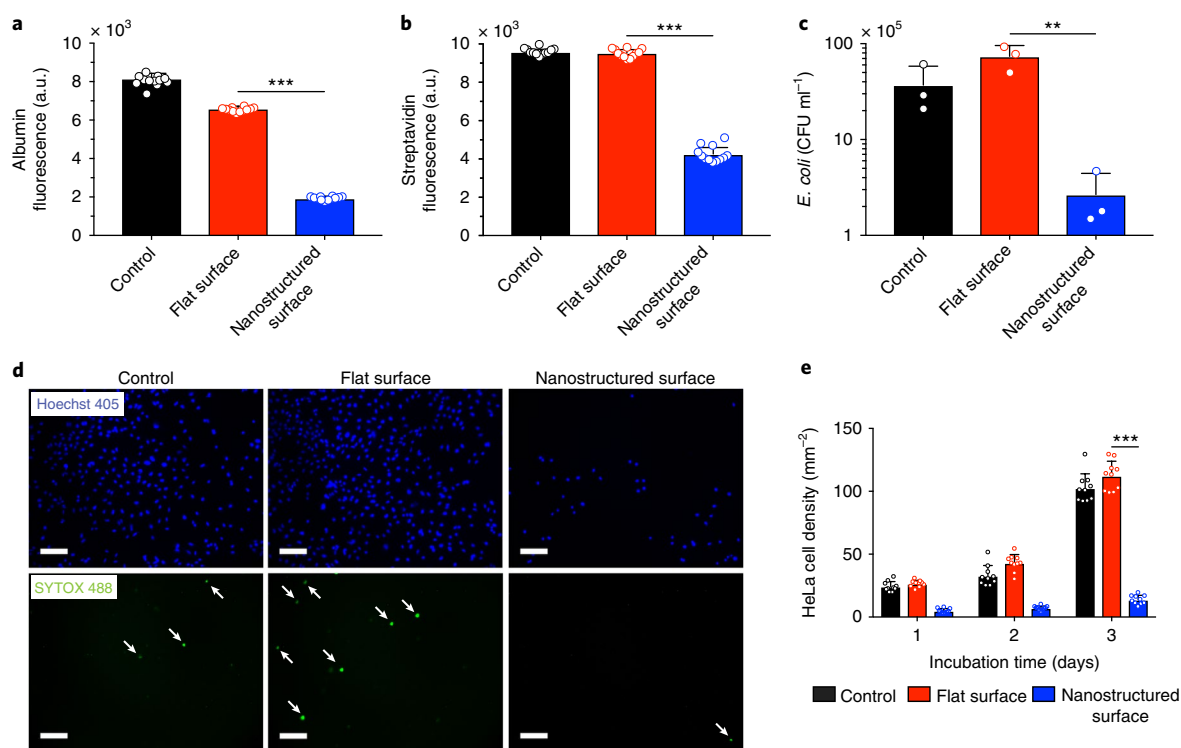


Fig. 3 | Nanostructured Si₃N₄ surface biophysical properties. **a, b**, Adhesion force characterized using fluorescence-intensity microscopy for bovine serum albumin (**a**) and streptavidin (**b**) on positive control, flat Si₃N₄ and nanostructured Si₃N₄ surfaces. Nanostructured Si₃N₄ surfaces show significant reduction in albumin and streptavidin adhesion relative to the control and flat Si₃N₄ (****P* < 0.001, one-way ANOVA with post-hoc Tukey test, error bars show s.d., *n* = 12 representative images). **c**, The number of adherent CFUs of *E. Coli* on the nanostructured Si₃N₄ surface was significantly lower than that on the flat Si₃N₄ surface (***P* < 0.01, one-way ANOVA with post-hoc Tukey test, error bars show s.d., *n* = 3 agar plates). **d**, Fluorescent micrographs of a positive control, flat Si₃N₄ and nanostructured Si₃N₄ incubated for 72 h in HeLa cell cultures labelled with cell-permeable nucleic acid markers Hoechst 405 (upper panels) and SYTOX Green (lower panels) indicating the anti-adhesive properties of nanostructured Si₃N₄. The arrows in the micrographs indicate dead cells. Scale bars, 100 μm. **e**, Adherent HeLa cell density on the nanostructured Si₃N₄ surface was significantly lower than on the positive control and the flat Si₃N₄ surface (****P* < 0.001, two-way ANOVA with Bonferroni's multiple comparisons test, error bars show s.d., *n* = 10 representative images). Adjustments were made for multiple comparisons for all statistical tests used. Experiments **a** and **b** were replicated three times whereas **c–e** were replicated twice.

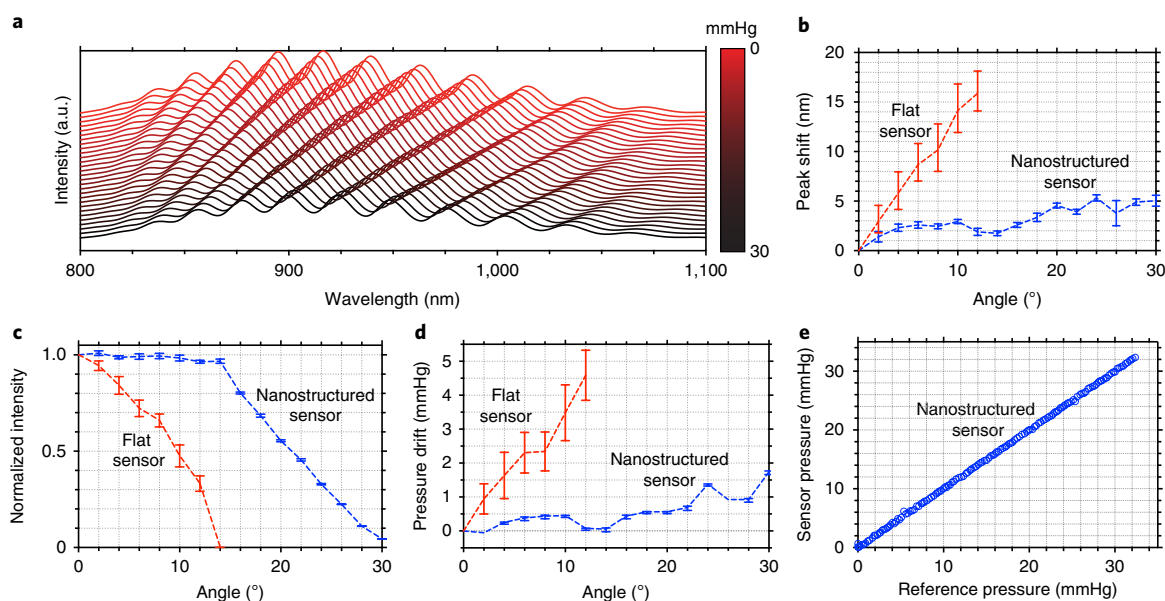


Fig. 4 | Benchtop characterization of nanostructured IOP sensor. **a**, Resonance shifts of the sensor Fabry-Perot cavity measured in reflection as a function of the IOP. **b**, Peak shift in the reflected resonance spectra as a function of incident angles ($n=3$ measurements). Considerably smaller magnitudes of peak shifts are observed in the nanostructured sensor, indicating its angle-independent property. **c**, Intensity, taken as a measure of peak-to-valley contrast of the most prominent peak and valley of the resonance profile and normalized with respect to the measurement taken at a 0° angle of incidence. The nanostructured sensor displays negligible loss of intensity up to 14° ($n=3$ measurements). **d**, Pressure drift error induced by increasing the angle of incidence. The nanostructured sensor displays negligible pressure drifts even at large angles of incidence ($n=3$ measurements). **e**, Nanostructured sensor tested from 0–32 mmHg in a pressure-controlled chamber interfaced with a digital pressure gauge used as a reference. Error bars show the s.d. about the mean. All the experiments were conducted once.

promote autogenous lysis when placed in contact with mammalian cells⁴⁴. Such anti-biofouling approaches relying on physical lysis could undesirably damage tissues surrounding implants and elicit inflammation. Supplementary Table 1 shows physical lysis occurs on either natural or synthetic nanostructured surfaces if the aspect ratio of the nanostructures is one or greater. Hence, by keeping the aspect ratio of the nanostructures at 0.45, the anti-adhesion property was leveraged to prevent biofouling without causing any physical lysis. Additionally, the hydrophilicity of the nanostructured surface originates from surface topology, which may provide better long-term reliability over chemical-treatment methods. (See Supplementary Section 3 and Supplementary Fig. 17.)

Use of nanostructures in intraocular pressure sensing

To demonstrate a medical application for multifunctional nanostructures, we used the nanostructured Si_3N_4 -membrane as an opto-mechanical sensing element in a microscale implantable IOP sensor, which is a hermetically sealed, pressure-sensitive, Fabry-Perot resonator¹⁵. A flat-surfaced or nanostructured flexible Si_3N_4 -membrane forms the top surface of the Fabry-Perot resonator and a mirror-like rigid Si forms the bottom surface. The sensor is optimized in the NIR range for minimum absorption in tissue and water. If the ambient pressure or IOP changes, the membrane will deflect accordingly, and the resulting shift in the resonance wavelength will be captured remotely in reflection (Fig. 4a, Supplementary Fig. 19). In vivo testing showed that Fabry-Perot-based IOP sensors suffer from narrow readout angles that severely limit their practicality as sensors (Supplementary Fig. 20) and biofouling that shortens sensor lifespan¹⁵.

To study the dependence on readout angle, we compared the measurements from a nanostructured and a flat-surfaced IOP sensor at 1 atm (Fig. 4b–d). The flat-surfaced sensor produced a maximum resonance shift of 16 nm at an incident angle of 12° (Fig. 4b). By contrast, the nanostructured sensor produced shifts

of 2 nm at 12° and 5 nm at 30° . Decay in the intensity of reflected resonance was also measured as a function of the incident angle (Fig. 4c). For the flat-surfaced sensor, the intensity decayed to zero when the incident angle reached 12° while the signal from the nanostructured sensor remained detectable until 30° . The IOP-measurement error of the flat-surfaced sensor reached 4.59 mmHg at 12° (Fig. 4d), which is approximately 46% of the physiological IOP range observed in humans (10–20 mmHg) and exceeds the ± 1.2 mmHg error range of existing clinical tonometers (<http://www.icaretonometer.com/>, <http://www.reichert.com/>). On the other hand, the IOP-measurement error of the nanostructured sensor was 0.07 and 0.92 mmHg at 12° and 28° , respectively. These results highlight the wide-angle performance of the nanostructured sensor. The nanostructured sensor showed excellent linearity (correlation factor: ~ 1.00) over the clinical range of interest from 0 to 32 mmHg when tested in a pressure-controlled chamber interfaced with a digital pressure gauge (Fig. 4e). The maximum readout error was 0.26 mmHg, approximately four times lower than that of the flat-surfaced sensor (1 mmHg).

A nanostructured and a flat-surfaced sensor were implanted individually inside the anterior chambers of two New Zealand white rabbits to investigate in vivo optical performance and biocompatibility (Fig. 5a). To examine the stability of sensor measurements, the shift $\Delta\lambda$ of the most prominent peak in each spectrum of the set was computed with respect to the mean of the set (Fig. 5b). The s.d. of $\Delta\lambda$ of the nanostructured sensor was 0.6 nm as opposed to 1.3 nm observed for the flat-surfaced sensor (Fig. 5c). Additionally, the s.d. of IOP measurements produced using the nanostructured sensor was 0.23 mmHg as opposed to 0.64 and 1.97 mmHg calculated from measurements concurrently obtained using the flat-surfaced sensor and tonometry, respectively (Fig. 5d). The angle independence enhanced by the nanostructure integration improved the stability and accuracy of the optical measurements against potential error sources such as respiratory movements, subtle eye motions and

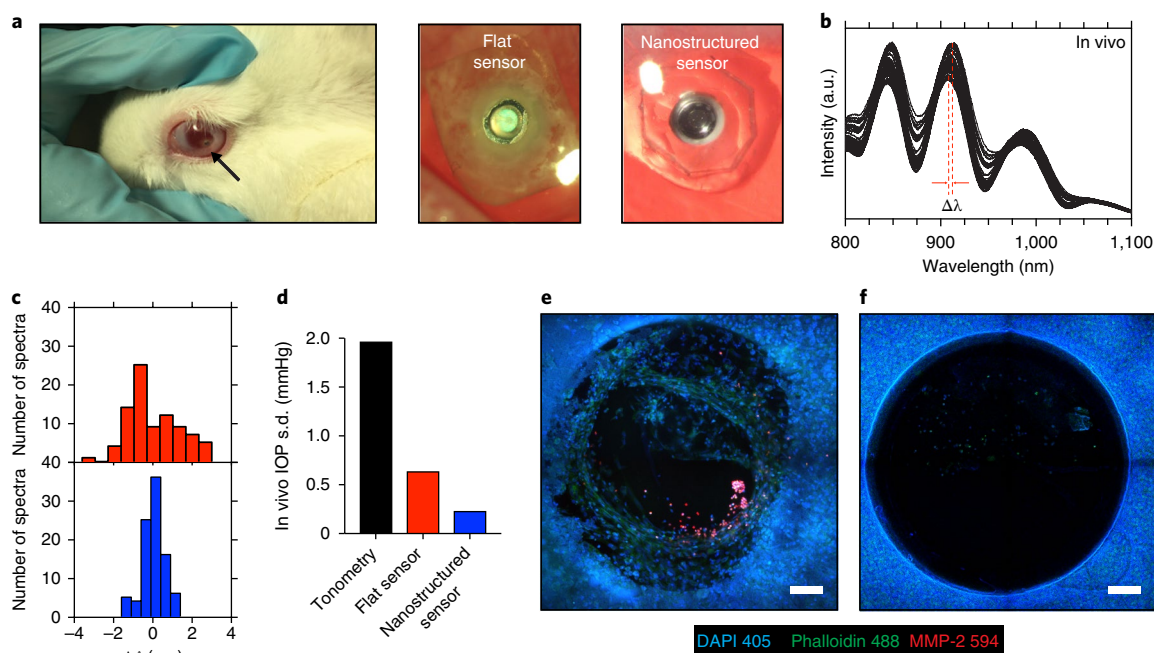


Fig. 5 | The in vivo performance and biocompatibility of the nanostructured IOP sensor. **a**, A flat-surfaced and a nanostructured sensor were each implanted in the anterior chamber (indicated by the black arrow) of two living New Zealand white rabbits for a period of one month. The sensors were mounted on a flexible silicone haptic and implanted through a minimally invasive surgical process¹⁵. The flat-surfaced sensor appears hazy compared with the nanostructured sensor due to dense tissue growth on the sensor surface indicating significant biofouling after one month of implantation. **b**, Spectra with the highest signal-to-noise ratio collected from continual IOP measurements taken over 60-s intervals with an integration time of 10 ms per spectrum. The variation in the position of the resonance spectra that occurred during a single set of measurements is indicated as $\Delta\lambda$ ($n = 95$ spectra). **c**, Histograms showing the numbers of spectra at specific $\Delta\lambda$ relative to the mean wavelength for the flat (s.d. = 1.3 nm) and nanostructured (s.d. = 0.6 nm) sensors, respectively ($n = 95$ spectra). **d**, Standard deviation of in vivo IOP measurements made using the flat (s.d. = 0.64 mmHg, $n = 95$ spectra) and nanostructured (s.d. = 0.23 mmHg, $n = 95$ spectra) sensors compared with a traditional rebound tonometry reading (s.d. = 1.97 mmHg, $n = 12$ measurements). **e**, Three-channel immunofluorescence confocal microscopy image (z-stack) of the flat Si_3N_4 sensor after one month of in vivo study. Several signs of inflammation (shown in red) over the flat Si_3N_4 -membrane of the sensor have elicited a foreign body reaction by means of a vast cellular migration process. Healthy tissue growth with a vast F-actin network is observed over the flat Si_3N_4 -membrane. (DAPI: cell nucleus marker, blue; phalloidin: cell F-actin marker, green; and MMP-2: matrix metalloproteinases marker, red.) **f**, Immunofluorescence image of a nanostructured sensor after one month in vivo study. Considerably reduced tissue adhesion over the nanostructured Si_3N_4 -membrane (the circular region) indicating the contribution of nanostructures to in vivo antifouling through anti-adhesion. Scale bars, 100 μm . Experiments **b–d** were replicated twice whereas **e** and **f** were conducted once.

detector misalignment. Furthermore, indirect IOP measurement techniques, such as tonometry, are influenced by various factors such as corneal thickness, curvature and biomechanics and are in general more error-prone compared with direct IOP measurement techniques such as implantable sensors^{15,45}.

Both sensors were retrieved after one month of implantation to quantify cell growth on the surface and to assess biocompatibility. We used confocal fluorescence microscopy to determine the extent of tissue growth and cellular viability at the time of retrieval. DAPI was used to localize all constituent cells while phalloidin, which selectively binds to actin, was used as an indicator of cellular processes and health⁴⁶. Additionally, matrix metalloproteinases-2 (MMP-2) was used as an indicator of inflammation for its role in various inflammatory and repair processes⁴⁷.

Figure 5e,f shows top views of the z-stacked multi-channel immunofluorescence images of the flat-surfaced and the nanostructured sensors, respectively. Approximately 59% of the flat-surfaced sensor was covered by tissue, and the presence of a vast filamentous F-actin network (Fig. 5e, in green) indicates healthy tissue growth at the time of extraction. Additionally, MMP-2 (Fig. 5e, in red) was observed over the membrane of the flat-surfaced sensor, which could have triggered the extensive cell migration towards this region. In comparison, approximately 5% of the nanostructured surface was covered by tissue, which was a 12-fold improvement

over the flat-surfaced sensor, and there was no detectable MMP-2 signal, suggesting the cell signalling and migration patterns present on the flat-surfaced sensor were absent on the nanostructured sensor. This indicates no inflammation occurred post-implantation and highlights the promising role of the nanostructures towards significantly improving in vivo biocompatibility of medical implants.

Conclusions

Inspired by the short-range-ordered nanostructures found on the wings of the transparent longtail glasswing butterfly (*C. faunus*), we engineered biophotonic nanostructures optimized for use in medical implants. By tuning key physical dimensions of the nanostructures, we engineered structurally induced scattering that expands optical readout angle and improves antifouling with suppressed inflammation suitable for IOP-sensing implants. In glaucoma, accurate IOP monitoring is the only mainstay of disease diagnosis and management⁴⁵, and optical sensing approaches for IOP monitoring have been promising in terms of miniaturization, energy efficiency and frequency of monitoring¹⁵; however, they also require improvement in readout angle and biocompatibility for practical use. Integration of the nanostructures on an IOP-sensing implant significantly expanded its detection range while reducing three-fold the mean in vivo IOP error. Further, the nanostructures effectively suppressed biofouling and inflammation 12-fold, resulting in

a highly practical implant for long-term IOP monitoring. Further development of our bio-inspired work, including continuous IOP monitoring using mobile devices with integration of features such as memory-based tracking⁴⁸, will improve glaucoma treatment outcomes and lower the risk of visual impairment and blindness. With these promising results, we envisage numerous medical technologies and devices will benefit greatly from the multifunctionality of biophotonic nanostructures.

Methods

Methods, including statements of data availability and any associated accession codes and references, are available at <https://doi.org/10.1038/s41565-018-0111-5>.

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Author contributions

V.N., R.H.S. and H.C. conceived the study. V.N. and R.H.S. designed the analyses while supervised by H.C. R.H.S. conducted the microscopy and spectroscopy of the longtail glasswing butterfly. R.H.S. conducted the simulations and numerical analysis. V.N. and R.H.S. fabricated and characterized the nanostructured Si_3N_4 -membrane samples. V.N., R.H.S., S.K. and N.H. conducted the *in vitro* tests. V.N., J.L. and R.H.S. fabricated

and characterized the benchtop IOP sensors. V.N., J.L. and J.D. performed the in vivo experiments under the supervision of D.S., V.N. and B.N. conducted the biocompatibility experiments of the in vivo IOP sensors. V.N., R.H.S. and H.C. co-wrote the manuscript with assistance from D.S. All authors discussed the results and commented on the manuscript.

Competing interests

The authors declare no competing interests.

Additional information

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Methods

High-resolution imaging. Dried wings of *C. faunus*, purchased from Bicbubs LLC (Arizona, USA) were coated with a 15 nm gold layer (Lesker Labline E-beam Evaporator, Kurt J. Lesker) before examination by SEM (FEI Nova 200 NanoLab Dualbeam) operated at 5 kV.

Topographical analysis. ImageJ (<https://imagej.nih.gov/ij/>), a public domain and Java-based image processing tool, was used for statistical analysis of nanostructure size on the wing membrane and fabricated nanostructured Si_3N_4 samples. Every pixel in the images was turned into either black or white using a threshold value that was obtained by calculating the mean intensity value of all the pixels of the same image. Subsequently, the diameters and areas of the nanostructures were computed. The 2D Fourier power spectra were obtained from SEM images and calculated with a fast Fourier transform algorithm in MATLAB.

Optical simulation. The thin-film simulations of flat Si_3N_4 -membrane and the postdiscal area were calculated using MATLAB[®]. The developed multilayer thin-film calculator is based on matching the boundary conditions for Maxwell's equations. Calculations were first done for individual polarizations and by taking the average afterwards to take the unpolarized light $((\text{TE} + \text{TM})/2)$, where TE and TM stand for transverse electric and transverse magnetic polarization components, respectively) condition into account.

The optical properties of the basal area nanostructures were numerically simulated with the 2D finite-difference time-domain software (Lumerical Solutions, Canada). The transmittance of nanostructured membrane was numerically simulated with the 3D finite element method (FEM) (COMSOL Multiphysics). To simulate the exact optical properties of the fabricated nanostructured samples, the measured 3D patterning profile was directly imported from the SEM in the simulated model. Periodic boundary conditions in the lateral directions (x and y) were applied for the calculations. An incoming plane wave impinging on the structures under normal and oblique incidence was used and all calculations were performed with a spatial resolution of 5 nm. In the simulation process, the boundary conditions of the electromagnetic fields in the vertical (z) direction were set on the perfect matching layer (PML) for the model. The total transmittance was calculated for individual geometries for unpolarized light $((\text{TE} + \text{TM})/2)$ at normal and oblique incident angles by integrating the near-zone scattered power (Poynting vector) over a surface before the bottom PML and afterwards normalizing with incident intensity.

Optical spectroscopic analysis. Specular transmission and scattering spectra of the *C. faunus* wings were measured using a custom-built optical goniometric set-up. A stabilized tungsten-halogen light source (SLS201, Thorlabs) was collimated to form a 500- μm -wide parallel incident beam that illuminates the sample at a fixed angle. The specular transmission and forward scattered light was detected at fixed and different angles, respectively, with an angular resolution of 2° and coupled into an optical fibre connected to the spectrometer (Flame, Ocean Optics). All measurements were recorded with unpolarized light.

The diffuse transmittance measurements were performed using a commercial Cary 5000 UV-Vis-NIR with integrating sphere. All measurements were recorded with unpolarized light. The samples were placed in the middle of the integrating sphere using a vice-type centre-mount and the sample holder was rotated around the vertical axis for angle-resolved measurements. Transmission measurements were normalized to that of the uncovered area of the underlying glass slide.

Phase-separation through polymer blends and nanostructure texturing. Poly(methyl methacrylate) (PMMA, $M_w = 9,590$, Polymer Standards Service GmbH) and polystyrene (PS, $M_w = 19,100$, Polymer Standards Service GmbH) were dissolved in methyl ethyl ketone (MEK, Sigma-Aldrich) with a mass ratio of 65% and 35%. The concentrations of the solutions were fixed at 25 mg ml^{-1} . Solutions were spin-coated on the substrates with a speed of 3,500 r.p.m. and acceleration of 2,000 r.p.m. s^{-1} for 30 s. Relative humidity was maintained between 40% and 50% during the spin coating. The de-mixing of the blend components out of the smooth surface occurred during spin coating itself due to the difference in relative solubility of PS and PMMA in MEK²⁸. When the sample begins to spin, water condensation begins at humidity levels above 35%. A layer of water-rich solution is formed at the air/solution interface due to the difference in evaporation rate between water and MEK. Water starts to condense from the air into the solution because of the evaporation of MEK, which decreases the temperature on top below the dew point. Because of the high water concentration, a 3D phase separation occurs between PS/MEK and PMMA/MEK/water. When the film is completely dried, a purely lateral morphology is formed and the PS islands end with an ellipsoidal shape. The samples were then rinsed in cyclohexane for 2 min and dried in a stream of N_2 to remove the PS islands. Using the PMMA layer as a template, a 30 nm Al_2O_3 hard masking layer was deposited via e-beam evaporation (CHA Industries Mark 40). After lift-off, the Si_3N_4 was textured through reactive ion etching (RIE, Oxford PlasmaLab 100 ICP380) resulting in the nanostructured surface.

Nanostructured membrane and sensor fabrication. The fabrication process flow of the nanostructured membrane and IOP sensor is provided in Supplementary Fig. 18.

The sensor consists of two parts (top and bottom), which were individually batch fabricated (T1–T5 and B1–B5, respectively in Supplementary Fig. 18) and bonded together using a medical grade epoxy to produce a hermetically sealed miniaturized Fabry–Perot cavity. (T1) Fabrication of the top substrate begins with 2- μm -thick SiO_2 and 400-nm-thick Si_3N_4 layers deposited on the top and bottom surfaces of a double-side-polished (DSP) Si wafer (thickness: 300 μm) using thermal oxidation and low-pressure chemical vapour deposition (LPCVD), respectively. (T2) The Si_3N_4 and SiO_2 layers on the top surface of the wafer were completely removed using RIE (Plasmalab System 100 RIE/ICP, Oxford Instruments) and buffered oxide etch (BOE). Next, a 300-nm-thick Al oxide (Al_2O_3) layer was deposited using an e-beam evaporator (FC-1800 E-Beam Evaporator, Temescal) and patterned the surface using photolithography and BOE. (T3) Using the patterned Al_2O_3 layer as a hard mask, the wafer was etched using a Bosch process (Plasmalab System 100 RIE/ICP, Oxford Instruments) down to the SiO_2 etch stop at the bottom surface of the wafer. (T4) The SiO_2 layer was removed through BOE to create freestanding Si_3N_4 -membranes. The nanostructuring process described in the previous section was used to structure the Si_3N_4 -membrane. (T5) Individual nanostructured membranes were released from the substrate through photolithography and RIE from the backside. (B1) The bottom substrate fabrication begins with a DSP Si wafer (thickness: 300 μm). (B2) Using a photoresist mask, a precisely controlled 4 μm recess was created through RIE to generate the Fabry–Perot cavity gap. (B3) A 300-nm-thick Al_2O_3 layer was deposited using an e-beam evaporator and patterned. (B4) Using a single Bosch process, concentric shallow trenches and a deep trench were etched. The shallow trenches were created to serve as reservoirs during the epoxy bonding process to prevent any of the adhesive from overflowing into the Fabry–Perot cavity. The deep trenches were created for the easy separation of bottom chips (bottom parts of the sensors) from the wafer. (B5) The Al_2O_3 masking layer was removed in BOE. Finally, a medical grade epoxy was applied along the sides of the top nanostructured membrane chip and the bottom chip to create a hermetically sealed Fabry–Perot sensor implant.

Nanostructured sensor characterization. The sensors were placed on a tilt stage, which allows for variation of the incident angle, and were probed by a $\times 20$ objective lens interfaced with an NIR light source (HL-2000, Ocean Optics), a mini-spectrometer (Maya200 Pro, Ocean Optics) and a CCD camera (Thorlabs). For linearity measurements, the sensors were placed in a custom-build pressure-controlled chamber. The hydrostatic pressure was between 0 to 32 mmHg and increased in steps of 0.2 mmHg. The output from the sensors were referenced against a digital pressure gauge (1210 Pressure Sensor, TE Connectivity) with an accuracy of $\pm 0.5\%$. An in-house IOP detection algorithm on MATLAB was used to rapidly calculate the IOP based on the peak locations of the captured reflection spectra (Supplementary Fig. 15).

Protein adhesion assay. Fluorescently labelled bovine serum albumin (A13100, Thermo Fisher Scientific) protein with a molecular weight of 66 kDa and streptavidin (S21375, Thermo Fisher Scientific) were used for the protein adsorption studies of the control, flat and nanostructured Si_3N_4 samples. The BSA and streptavidin were dissolved separately in phosphate buffered saline (PBS, 10 mM, pH 7.4) to a concentration of 2 mM. The substrates were rinsed with PBS to rehydrate the surfaces. All the sample substrates were then immersed in both protein solutions separately and were incubated at 4°C for 24 h. The samples were then removed from the protein solutions, gently washed three times with PBS, and rinsed once with deionized water to remove the PBS salt. Surface protein adsorption was imaged using a Hamamatsu ORCA-Flash4.0 V2 Digital CMOS camera on a Nikon Eclipse 80i fluorescence microscope with a $\times 10$ objective. ImageJ/FIJI (<https://imagej.nih.gov/ij/>) was used to quantify the protein adsorption data on 12 different imaging areas from each sample. All images were converted into binary images with a fixed threshold to enable sample comparison. Statistical methods used to analyse the data were obtained using Prism (GraphPad Software).

Prokaryote adhesion assay. A culture of chemically competent *E. coli* (Edvotek) was used directly after purchase and transformed with the green fluorescent protein expressing and ampicillin-resistant plasmid pFluoroGreen (Edvotek) followed by plating on selective agar for overnight incubation at 37°C . The bacteria were then inoculated in fresh lysogeny broth (LB) with $100 \mu\text{g ml}^{-1}$ ampicillin and isopropyl- β -D-thiogalactopyranoside (IPTG) and incubated overnight at 37°C under gentle shaking (250 r.p.m.). On recovery, the bacteria were diluted in fresh LB to an OD_{600} of 0.25. Two sets of autoclaved substrates consisting of positive control, flat Si_3N_4 and nanostructured Si_3N_4 were incubated with 2 ml of the bacterial culture. The sets were used for fluorescence imaging and CFU counts. After 4 h incubation under gentle shaking (100 r.p.m.), the substrates were washed with $1 \times$ PBS and placed in a sonication bath for 15 min. The substrates for fluorescence intensity measurements were analysed using a wide-field epifluorescence microscope ($\times 10$ objective, Leica DMI 600, Leica AG). For CFU counts, a 10^{-5} serial dilution was performed for the control and flat Si_3N_4 while a 10^{-4} serial dilution was performed for the nanostructured Si_3N_4 . The bacterial suspension from the substrates were then plated onto three selective agar plates per substrate. Statistical methods used to analyse the data were obtained using Prism (GraphPad Software).

Eukaryote adhesion assay. Three sets of autoclaved substrates consisting of positive control, flat Si_3N_4 and nanostructured Si_3N_4 were incubated in complete Dulbecco's modified Eagle's medium (DMEM) (10% fetal bovine serum (FBS) and 1% 100× penicillin/streptomycin) at 37°C and 5% CO_2 for 30 min. The medium was then aspirated followed by re-addition of fresh complete DMEM. HeLa cells (ATCC, maintained at low passages to avoid contamination) at a fixed concentration of $250,000\text{ ml}^{-1}$ were seeded on each substrate and measurements were collected at 24-h time point measurements, three in total. At each 24-h time point, one set of substrates was incubated in a staining reagent consisting of fresh complete FluoroBrite DMEM (10% FBS, 1% 100× penicillin/streptomycin, 1:1,000 Hoechst 33342, and 1:1,000 SYTOX Green) at 37°C and 5% CO_2 for 15 min. Hoechst 33342 (excitation/emission: ~350/461 nm) and SYTOX Green (excitation/emission: ~554/567 nm) probes were used to stain the nuclei of adherent HeLa cells on the surfaces under test for live imaging (Fig. 2i). Hoechst 33342 is nonspecific for either dead or live cells and provides an estimate of total adherent cell density, whereas SYTOX Green is impermeant into live cells and functions as an indicator of cell death. The co-localization of the two nucleic acid markers was used to ascertain mortality ratios (dead/live) for each surface. For the total adherent cell density measurement, 10 representative images were obtained through wide-field epifluorescence microscopy (×10 objective, 2 mm diameter of field area, Leica AG) by scanning a fixed $2\text{ cm} \times 2\text{ cm}$ substrate of each surface. Using a fixed field area and representative images, an average adherent cell density (count per mm^2) was obtained for each surface. An estimate of viability was computed as an average ratio of number of dead cells and live cells per field-of-view taken over 10 representative images. The co-localization of the two labels yielded the number of dead cells per representative field-of-view. The number of live cells was obtained by subtracting the number of dead cells from the total cell count tagged by Hoechst 33342 alone. The ImageJ/FIJI software was used to perform all required measurements. Statistical methods used to analyse the data were obtained using Prism (GraphPad Software).

In vivo IOP measurements. The sensors, mounted on silicone haptics, were folded and inserted into the anterior chamber through a 2.8 mm corneal incision¹⁵. On spontaneous unfolding, the haptics were positioned into the iridocorneal angles. A custom-built hand-held detection system was used for the in vivo IOP measurements from the nanostructured IOP sensor. The system was interfaced with a high-resolution NIR mini-spectrometer (Maya200 Pro, Ocean Optics) and a portable tungsten-halogen light source (HL-2000, Ocean Optics). To obtain a single IOP measurement, spectra were captured for a 60 s period with an integration time of 10 ms per spectrum, resulting in 6,000 spectra. Many of these measurements are influenced by naturally induced respiratory and subtle eye motions of the rabbits, which cause the angle of incidence to deviate from normal and increases error. Hence, 95 representative reflection spectra with the highest signal-to-noise ratio were chosen out of 6,000 spectra and used to calculate

the IOP. An in-house IOP detection algorithm on MATLAB was used to rapidly calculate the IOP based on the peak locations of the captured reflection spectra (Supplementary Fig. 15). Rebound tonometry was performed using a hand-held off-the-shelf system (TonoVet).

Confocal immunofluorescence microscopy sample preparation. Once harvested, the sensors were rinsed in $1 \times \text{PBS}$ with 0.02% NaN_3 and incubated in 4% paraformaldehyde (PFA) for 30 min at room temperature (RT) followed by overnight incubation at 4°C. Then, the sensors were rinsed in $1 \times \text{PBS}$ with 0.02% NaN_3 , followed by incubation in freshly prepared quenching buffer ($1 \times \text{PBS}$ with 0.02% NaN_3 and 50 mM NH_4Cl) for 15 min at RT. Following quenching, the sensors were rinsed again in $1 \times \text{PBS}$ with 0.02% NaN_3 . The sensors were then incubated in blocking buffer ($1 \times \text{PBS}$ with 0.02% NaN_3 , 2% BSA and 0.25% Triton X-100) overnight at RT with gentle shaking (20 r.p.m.). For staining, the sensors were incubated in 1 ml staining reagent containing blocking buffer with 1:1,000 DAPI 405 (cell nucleus marker), 1:500 phalloidin 488 (cell F-actin marker), and 1:500 MMP-2 594 (matrix metalloproteinases marker) overnight at RT with gentle shaking (20 r.p.m.) followed by 37°C for 1 h. Following staining, the sensors were washed in blocking buffer several times followed by incubation overnight at RT with gentle shaking (20 r.p.m.). All confocal imaging was performed with the sensors fully immersed in blocking buffer to retain the morphology of the tissue.

Confocal immunofluorescence microscopy and analyses. Imaging for in vivo biocompatibility analysis was performed using a laser scanning confocal microscope (LSM 880 with Airyscan, Carl Zeiss AG). The z-stack images (step size: $2.5\text{ }\mu\text{m}$, range: $500\text{ }\mu\text{m}$) were captured using a ×25 EPI objective with immersion oil, and controlled by Zeiss ZEN 2.1 software. A 2×2 tiling with 10% overlap section of each sensor field of view was captured and then stitched together using imaging software (Zeiss ZEN, Carl Zeiss AG). All subsequent image analyses were performed with ImageJ/FIJI software.

Ethics. All animals were treated in accordance with the Association for Research in Vision and Ophthalmology (ARVO) Statement for the Use of Animals in Ophthalmic and Vision Research. All animal research was conducted under protocols approved by the Institutional Animal Care and Use Committee of the University of California San Francisco (protocol no. AN110948).

Reporting Summary. Further information on experimental design is available in the Nature Research Reporting Summary linked to this article.

Data availability. The data that support the plots within this paper and other findings of this study are available from the corresponding authors upon reasonable request.

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► Experimental design

1. Sample size

Describe how sample size was determined.

No sample size calculations were performed. The sample size (n) of each experiment is provided in the corresponding figure captions in the main manuscript and supplementary information files. Sample sizes were chosen to support meaningful conclusions.

2. Data exclusions

Describe any data exclusions.

No data was excluded from the analyses.

3. Replication

Describe the measures taken to verify the reproducibility of the experimental findings.

All in vitro experiments were replicated successfully 2 or 3 times. In vivo optical IOP measurements were conducted on one sensor of each type and were replicated successfully 2 times while immunofluorescence microscopy on the retrieved sensors was conducted once.

4. Randomization

Describe how samples/organisms/participants were allocated into experimental groups.

In the reported experiments, each group consisted of identically engineered samples. The work does not involve participant groups. Therefore, randomization was not relevant the study.

5. Blinding

Describe whether the investigators were blinded to group allocation during data collection and/or analysis.

Methods for group allocation, data collection and all related analyses were predetermined. Furthermore, the work does not involve participant groups. Therefore, blinding was not relevant to the study.

Note: all in vivo studies must report how sample size was determined and whether blinding and randomization were used.

6. Statistical parameters

For all figures and tables that use statistical methods, confirm that the following items are present in relevant figure legends (or in the Methods section if additional space is needed).

n/a Confirmed

- ☐ ☒ The exact sample size (n) for each experimental group/condition, given as a discrete number and unit of measurement (animals, litters, cultures, etc.)
- ☐ ☒ A description of how samples were collected, noting whether measurements were taken from distinct samples or whether the same sample was measured repeatedly
- ☐ ☒ A statement indicating how many times each experiment was replicated
- ☐ ☒ The statistical test(s) used and whether they are one- or two-sided
Only common tests should be described solely by name; describe more complex techniques in the Methods section.
- ☒ ☐ A description of any assumptions or corrections, such as an adjustment for multiple comparisons
- ☐ ☒ Test values indicating whether an effect is present
Provide confidence intervals or give results of significance tests (e.g. P values) as exact values whenever appropriate and with effect sizes noted.
- ☐ ☒ A clear description of statistics including central tendency (e.g. median, mean) and variation (e.g. standard deviation, interquartile range)
- ☐ ☒ Clearly defined error bars in all relevant figure captions (with explicit mention of central tendency and variation)

See the web collection on [statistics for biologists](#) for further resources and guidance.

► Software

Policy information about [availability of computer code](#)

7. Software

Describe the software used to analyze the data in this study.

COMSOL 5.3, ImageJ 1.51, Lumerical 8.19.1416, Matlab R2016b, Prism Graphpad 7.0d, Zeiss ZEN 2.1

For manuscripts utilizing custom algorithms or software that are central to the paper but not yet described in the published literature, software must be made available to editors and reviewers upon request. We strongly encourage code deposition in a community repository (e.g. GitHub). *Nature Methods* [guidance for providing algorithms and software for publication](#) provides further information on this topic.

► Materials and reagents

Policy information about [availability of materials](#)

8. Materials availability

Indicate whether there are restrictions on availability of unique materials or if these materials are only available for distribution by a third party.

No unique materials were used.

9. Antibodies

Describe the antibodies used and how they were validated for use in the system under study (i.e. assay and species).

This work used Alexa Fluor 594 anti-MMP2 antibody, an anti-MMP2 antibody directly conjugated with the alexa fluor 594 dye (Cat.# 679904) at a 1:500 dilution. The dye was purchased from BioLegend Inc (San Diego, CA 92121), Clone # M6303D01, Lot #B212008 - "https://www.biolegend.com/de-de/products/alexa-fluor-594-anti-mmp2-antibody-12822".

10. Eukaryotic cell lines

a. State the source of each eukaryotic cell line used.

The HeLa cells used in this work were purchased from ATCC.

b. Describe the method of cell line authentication used.

None of the cell lines have been authenticated.

c. Report whether the cell lines were tested for mycoplasma contamination.

The cell lines were not tested for mycoplasma contamination.

d. If any of the cell lines used are listed in the database of commonly misidentified cell lines maintained by [ICLAC](#), provide a scientific rationale for their use.

No commonly misidentified cell lines were used.

► Animals and human research participants

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11. Description of research animals

Provide all relevant details on animals and/or animal-derived materials used in the study.

New Zealand white Rabbit, female, 2.5kg, 5 months old.

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12. Description of human research participants

Describe the covariate-relevant population characteristics of the human research participants.

The study did not involve human research participants.